FORECASTING OF VITRIFICATION TEMPERATURE FOR FLUOROPOLYMERS

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Abstract. Methods of calculating the Tg was tested on perfluoropolyoxaalkylene triazine polymers and used to calculate the Tg of carbochain copolymers of perfluoroalkyl vinyl ethers with vinylidene fluoride (tetrafluoroethylene). The methods of calculation will help to predict the properties of fluorinated polymers and Tg values for the new structures

Keywords. Copolymers, fluoropolymers, perfluoroalkyl vinyl ethers, triazine

It is well known that introduction of ether groups into the main or side chains of a polymer improves low-temperature properties of fluorelastomers and fluorocarbon oils [1].

Vitrification temperature (T_g) is known to be the major characteristic among low-temperature properties of amorphous polymers, particularly, elastomers; the said temperature is detected by various methods including those involved in the standard [2].

A number of methods proposed for the calculation of vitrification temperature [3, 4] were based on the polymer chemical structure reasoning from the idea that each constitutional repeating unit of the macromolecule under consideration should be divided into groups of atoms that contribute additive inputs into the equation for T_g independently of their surrounding. Those methods are not universal and their accuracy in fluoropolymer calculations never exceeded +/-10C⁰. The following equation for the calculation of ($Ig T_g$) was proposed by A.A. Askadsky and G.L. Slonimsky [5] reasoning from the analysis of polymer chains packing factors:

$$lgT_{c} = \frac{\sum K_{i}^{*}}{N \sum \Delta V_{i}} + A \quad (1)$$

where:

 $\sum_{i=1}^{K_{i}^{*}}$ polymer unit effective volume, that is an additive value (with the dimensionality of mole volume) related both to the packing factors and volume coefficient of the polymer body;

 $\sum \Delta V_i$ - Van der Waals volume of the polymer unit;

N - Avogadro number;

A- polymer linearity parameter 1,435.

The similar approach was used to calculate T_g for fuorine-containing polymers, especially perfluoroalkyleneazines [7], through the introduction of K_i^* increments corresponding to fluorine-containing groups. Numerical values of the related effective volumes were determined and vitrification temperatures were calculated for about 50 polymers.

Unfortunately, at present there are virtually no methods for the production of high-molecule fluorocarbon polymers with ether bonds in their main chains, except for some polymers producible through the polycondensation, particularly perfluorooxaalkyltriazines.

However, experimental T_g values for the most of such perfluoropolymers do not satisfy the said calculation schedule. In Table 1 there are T_g , values determined according to the state standard [2] for more than 40 perfluoropolyoxaalkylenetriazine polymers synthesized by us and structured as below:



 R_F - perfluoropolyoxaethylene- or perfluoropolyoxaisopropylene- unit in the main chain of a perfluoroalkylenetriazine polymer; R'_F - perfluoro-3-oxapropyl-, perfluoro-1-methyl-2-oxapentyl- and perfluoro-1,4-dimethyl-2,5-dioxaoctyl, and also polyoxadifluoromethylene pendant in triazine ring.

R _F in the main chain	R' _F pendant in the side chain		^{exp} Tg, ^o C		
1	2		3		
	CF ₃ -		CF ₃ -		-15,5
- (CF ₂) ₆ -	CF ₃ OCF ₂ CF ₂ -		-41,5		
		3	-83,0		
	$CF_3O(CF_2O)_nCF_2$ -, where n=		-92,0		
		5	-105,0		
		6	-119,0		

Table 1. Vitrification temperatures for perfluorotriazine polymers

1			
	$CF_3(CF_2)_2O(CFCF_2O)_nCF,$ CF_3 CF_3	0	-24,0
	where n=	1	-36,0
		2	-41,5
- $CF_2O(CF_2)_3OCF_2CF_2OCF_2_$	CF ₃ -		-65,5
	CF ₃ OCF ₂ CF ₂ -		-75,0
	CF ₃ -		-75,0
	CF ₃ OCF ₂ CF ₂ -		-80,0
$-CFO(CF_2)_3OCF - CF_3 CF_3$	CF ₃ O(CF ₂ O) ₅ CF ₂ -		-103,0
$-CFO(CF_2)_6OCF - CF_3 CF_3$	CF ₃ OCF ₂ CF ₂ -		-45,0
$-CFO(CF_2)_3OCFCF_2CF$ CF_3 CF_3 CF_3	CF ₃ OCF ₂ CF ₂ -	-46,0	
-CFO(CF ₂) ₆ OCFCF ₂ CF- CF ₃ CF ₃ CF ₃		1	-57,0
	$CF_3O(CF_2O)_nCF_2$ -, where n=	3	-74,0
		5	-88,0
-CFOCF ₂ CF O(CF ₂) ₃ OCFCF ₂ OCF- CF ₃ CF ₃ CF ₃ CF ₃	CF ₃ -		-36,0
	CF ₃ OCF ₂ CF ₂ -		-48,0
	CF ₃ O(CF ₂ O) _n CF ₂ -, where n=	2	-66,0
		5	-89,0
		6	-95,0
		7	-96,0
		0	-38,5

1	CF3(CF2)2O(CFCF2O)CF-		F	-+	
	CF_3 CF_3	, wh	ere	1	-44,0
				2	-47,0
$- \underbrace{CFOCF_2CFO(CF_2)_3O(CFCF_2O)_2CF}_{CF_3} + \underbrace{CF_3} + \underbrace{CF_3}$	CF ₃ -				-41,0
		- n-		1	-57,5
	CF30(CF20)nCF2-, Where	: 11—		2	-65,0
				5	-84,0
				6	-89,0
				7	-90,5
	$CF_3(CF_2)_2OCFCF_2OCF-CF_3$ CF_3 CF_3				-45,5
	CF_3OCFCF_2OCF- CF_3 CF_3				-47,0
		1	-59,0		
$CF_3 CF_3 CF_3 CF_3 CF_3 CF_3 CF_3 CF_3 $	n= n=	2	-64,0		
		3	-68,0		
		5	-79,0		
		7	-90,5		
	CF ₃ O(CF ₂ CF ₂ O) ₃ CF ₂ -		-57,0		
	$CF_3(CF_2)_2OCFCF_2OCF - CF_3 CF_3$		-46,5		
	CF ₃ OCFCF ₂ OCF- CF ₃ CF ₃		-48,0		

The table analysis evidences that the introduction of fragments with ether oxygen into the main chain of a perfluorooxaalkylenetriazine polymer, especially due to tetrafluoroethyleneoxide units, decreases T_q considerably. In the case that oxygen is present in the chain due to hexafluoropropeneoxide units, the maximal effect of T_q decrease is achieved not only through the rise in number of oxygen-containing units in the main chain, but mostly through the rise in number difluorooxamethylene of units in side-chains R'_⊏. This enables, when using perfluoropolyoxamethylene side chain groups in those polymers (in order to decrease T_{α}), to decrease the number of hexafluoromethylenoxide units between triazine rings.

Therefore, one may conclude about multidirectional effects of oxygen-containing units in side and main chains, and that the maximal influence on the vitrification temperature is provided by oxygen of perfluoropolyoxamethylene groups in side pendants.

In this connection in the equation for the calculation of vitrification temperature two different increment values are to be used for oxygen's in main and side chains, those being K^*o_{main} and K^*o_{side} . It is obvious that $K^*o_{main} > K^*o_{side}$. For each of those coefficients the effects of the number of diperfluoromethylene or methyl groups located in the oxygen α - and b-surroundings in C_n -O-C_m should be taken into account as an average:

$$K^* o_{nn} = \frac{K^* / n + K^* / m}{2} \quad (2)$$

It is obvious that the effects of triazine ring on vitrification temperature should be taken into account by the introduction of a special coefficient K^*_{tr} .

Reasoning from experimental T_g values for perfluoroalkylenetriazine polymers in table 1 and $K^*_{i,i}$, used by A.A. Askadsky et al. [5, 6] the equation (1) was used to calculate the increment value K^*o_{side} for each oxygen in perfluoromethylene oxide units in side chains, $K^*_1o_{main}$ for each oxygen in the main polymer chain in hexafluoropropylene oxide units, $K^*_2o_{main}$ for oxygens in tetrafluoroethyleneoxide units and K^* tr for triazine ring, all shown in table 2.

Structure of the fragment	K [*] ı, cm ³ /mole	Structure of the fragment	K [*] ı, cm ³ /mole
CF ₃ OCF ₂ OCF ₂ O-× × ×	K*sub> <i>áîê</i> = -3,760	N	K*òð = 32,077
-CFOCF ₂ CF O CF ₃ CF ₃	K [*] 10 _{main} =1,324	Ň	

Table 2. Novel incremental values in equation (1)

- $CF_2OCF_2CF_2O-x \times x$ $K_2^*o_{main}=-1,220$			
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Using the novel increment values T_g were calculated for a number of perfluoroalkylenetriazine polymers, and thus obtained values are compared with experimental ones [2] in Table 3.

Table 3. Calculated and experimental values of vitrification temperatures forperfluoroalkylenetriazine polymers

-R _F in the main chain	R' _F in the side chain		T _g , ^o C		Δ ±°, _C
			exp.	calc.	
1	2		3	4	5
		1	-15,5	-15,5	0
- (CF ₂) ₆₋	$C_n F_{n+1}$ -, where n=	3	-23,0	-13,0	-10
		6	-26,0	-10,0	-16
	CF ₃ OCF ₂ CF ₂ -		-41,5	-38,0	-3,5
	CF ₃ O(CF ₂ CF ₂ O) ₃ CF ₂ -		-69,0	-64,5	-4,5
		1	-55,0	-59,0	4,0
	$CF_3O(CF_2O)_nCF_2^-,$ where n=	2	-68,0	-74,0	6,0
		3	-83,0	-86,0	3,0
		4	-92,0	-95,0	3,0
		5	-105,0	-102,5	-2,5
		6	-111,0	-108,5	-2,5
		7	-119,	-114	-5,0
	$CF_3(CF_2)_2O(CFCF_2)_nOCF$ CF_3 CF_3	1	-24,0	-27,0	3,0

	where n=	2	-36,0	-36,0	0
		3	-41,5	-41,5	0
	CF_3OCFCF_2OCF CF_3 CF_3		-39,0	-43,0	3,0
- $CF_2O(CF_2)_3OCF_2CF_2OCF_2_$	CF ₃ -		-65,5	-65,0	-0,5
	CF ₃ OCF ₂ CF ₂ -		-75,0	-76,5	1,5
	CF ₃ -		-75,0	-73,0	-2,0
	CF ₃ OCF ₂ CF ₂ -		-80,0	-78,5	-1,5
$-CFO(CF_2)_3OCF - CF_3 CF_3$	CF ₃ O(CF ₂ O) _n CF ₂ -,	3	-82,0	-88,5	6,5
	where h=	5	-103,0	-103,0	0
-CEO(CE), OCECE, CE-	C _n F _{n+1} -, where n=		-32,5	36,0	3,5
$-CFO(CF_2)_3OCFCF_2CF - CF_3 CF_3 CF_3 CF_3$			-39,0	31,5	-7,5
	CF ₃ OCF ₂ CF ₂ -		-46,0	-46,0	0
	CF ₃ O(CF ₂ O) _n CF ₂ -,	1	-57,0	-53,5	-3,5
	wnere n=	3	-82,0	-88,5	6,5
		5	-88,0	-86,0	-2,0
$\begin{array}{c} -CFO(CF_2)_6 \\ CF_3 \end{array}$	CF ₃ OCF ₂ CF ₂ -		-48,0	-40,0	-4,0
	CF ₃ OCF ₂ OCF ₂ -		-57,0	53,5	-3,5
-CEOCE-CEOCE-	C _n F _{n+1} -, where n=	1	-36,0	-37,5	-1,5
CF_3 CF_3 CF_3 CF_3 CF_3		3	-43,5	-34,0	-9,5
		6	-41,0	-29,5	-12,5

1					
	CF ₃ OCF ₂ CF ₂ -		-48,0	-47,5	-0,5
	CF ₃ O(CF ₂ O) _n CF ₂ -,	2	-66,0	-69,0	3,0
	where n=	4	-79,5	-83,5	4,0
		5	-89,0	-89,5	0,5
		6	-95,0	-95,0	0
	CF ₃ (CF ₂) ₂ O(CFCF ₂) _n OCF- CF ₃ CF ₃	1	-38,5	-40,0	1,5
	where n=	2	-44,0	-44,0	0
		3	-47,0	-47,0	0
	CF ₃ OCFCF ₂ OCF- CF ₃ CF ₃		-45,0	-49,0	4,0
	C _n F _{n+1} -, where n=	1	-41,0	-39,0	-2,0
CF_3 CF_3 CF_3 CF_3 CF_3		3	-47,5	-35,5	-12,0
		6	-43,0	-31,5	-11,5
	CF ₃ OCF ₂ CF ₂ -		-52,5	-48,0	-4,5
		1	-575	-57,5	0
	where n=	2	-640	-63,5	0,5
		3	-69,5	-73,0	3,5
		4	-75,0	-79,5	4,5
		5	-84,0	-85,0	1,0
		6	-89,0	-90,0	1,0
		7	-96,0	-94,5	-1,5

	$CF_3(CF_2)_2OCFCF_2OCF - CF_3 CF_3 CF_3$		-46,5	-44,0	-2,5
	CF_3OCFCF_2OCF CF_3 CF_3		-47,0	-48,5	1,5
	$C_n F_{n+1}$ -, where n=	1	-45,5	-40,0	-5,5
CF_3 CF_3 CF_3 CF_3 CF_3		3	-51,5	-37,0	-14,5
		6	-45,5	-33,0	-12,5
	CF ₃ OCF ₂ CF ₂ -		-52,5	-48,0	-4,5
		1	-59,0	-56,0	-3,0
	where n=	2	-64,0	-63,5	-0,5
		3	-68,0	-70.0	2,0
		4	-72,0	-76,0	4,0
		5	-79,0	-81,5	2,5
		6	-82,0	-86,0	4,0
		7	-90,5	-90,5	0
		8	-99,5	-94,0	-4,5
		9	-103	-97,5	-5,5
	CF ₃ OCFCF ₂ OCF- CF ₃ CF ₃		-48,0	-48,0	0
$-CFO(CF_2CFO)_2(CF_2)_3O(CFCF_2)_3CF- CF_3 CF_3 CF_3 CF_3 CF_3 CF_3 CF_3 CF_3$	CF ₃ (CF ₂) ₂ OCFCF ₂ OCF- CF ₃ CF ₃		-48,0	-44,0	-4,0
1	2	3	4	5	
	$C_n F_{n+1}$, where n= 1	-46,	-41,	0 -5,0	

-CFO(CF ₂ CFO) ₃ (CF ₂) ₃ O(CFCF ₂) ₃ CF- \CF ₃ \CF ₃ \CF ₃ \CF ₃ \CF ₃		3	-52,0	-38,5	-13,5
		6	-48,5	-35,5	-13
	CF ₃ O(CF ₂ O) ₆ CF ₂ -	6	-83,0	-80,0	-3,0
	CF ₃ OCFCF ₂ OCF- CF ₃ CF ₃		-51,0	-47,5	-3,5

Root mean square deviation of the calculated value from that experimental does not exceed +/-5,2°_C. To estimate the influence of oxygen in the main chain of hetero-chain co-polymers we, making use of K^*o_{nm} increments calculated according to (2), estimated T_g values for a number of polyethers and compared them in Table 4 with experimental values found in literature [6-8]. It is obvious that the limit polyether vitrification temperature value will be that calculated for polydifluoromethylene oxide -170,5°C as shown in the table.

-R _F in the main chain	-R _F in the main chain			$\Delta \pm ^{0}_{,C}$
		exper.	calc.	
- [- (CF ₂ CF ₂ O) ₃ (CF ₂ O) ₇₋] _{n-}		-160,0	-144,5	-15,5
- (-CF ₂ CF ₂ CF ₂ CF ₂ O-) _{n-}	-65,0	-52,0	-13,0
- (- CF ₂ CF ₂ O-) _{n-}		-95,0	-99,0	4,0
- (- CF ₂ O-) _{n-}		-	-170,5	
-R _F in the main chain	R' _F in the side chain			
-(-(F(F,O-) -*	CF ₃	-72,0	-68,5	-3,5
'R'F	CF ₃ O(CF ₂) ₃ O-		-73,5	
	CF ₃ OCF ₂ O(CF ₂) ₃ O-		-94,5	

Table 4. Calculated and experimental vitrification temperatures for perfluoropolyethers

	CF ₃ O(CF ₂) ₃ O(CF ₂) ₃ O-		-108,0	
	CF ₃ O(CF ₂) ₄ O(CF ₂) ₃ O-		-125,0	
	CF ₃ O(CF ₂) ₅ O(CF ₂) ₃ O-		-130,0	
	CF ₃ O(CF ₂) ₆ O(CF ₂) ₃ O-		-134,5	
	CF ₃ O(CF ₂) ₇ O(CF ₂) ₃ O-		-138,0	
-[(-CFCF₂O)₄CF₂O-L-	CF ₃		-80,0	
R'F	CF ₃ O(CF ₂) ₃ O-	-104,0	-99,0	-5,0
	CF ₃ OCF ₂ O(CF ₂) ₃ O-		-125,0	
	CF ₃ O(CF ₂ O) ₂ CF ₂ CF ₂ O-		-133,0	
	CF ₃ O(CF ₂ O) ₃ CF ₂ CF ₂ O-	-135,0	-138,5	3,5
	CF ₃ O(CF ₂ O) ₄ CF ₂ CF ₂ O-		-142,0	
	CF ₃ O(CF ₂ O) ₅ CF ₂ CF ₂ O-		-146,0	
	CF ₃ O(CF ₂ O) ₆ CF ₂ CF ₂ O-		-148,0	
	CF ₃ O(CF ₂ O) ₇ CF ₂ CF ₂ O-		-150,0	

* - to calculate T_g for polyhexafluoropropyleneoxide with such structure $K_2^*o_{main}$ increment was applied.

The method proposed for the calculation of vitrification temperature was tested with perfluoropolyoxaalkylenetriazine polymers and used to calculate T_g values for carbon-chain copolymers of perfluoroalkylvinyl ethers with vinylidenefluoride and tetrafluoroethylene produced through emulsion polymerization. In Table 5 thus calculated values are compared with those experimental.

Table 5. Calculated and experimental vitrification temperatures for carbon-chain fluoropolymers

$-(-CH_2CF_2)_3CFCF_2-)_n-*$				
R' _F in the side chain		T _{g,} ⁰ _C		$\Delta \pm ^{o}{}_{C}$
		exper.	calc.	
CF ₃ (CF ₂) _n O- , where n=	0	-40	-42,0	2,0
	1	-27	-29,5	2,5
	2	-33	-29,5	-3,5
$CF_3O(CF_2)_nO-$, where n=	2	-41,0	-56,5	15,5
	3	-50,0	-50,0	0
$CF_3O(CF_2CF_2)_nO-$, where n=	2	-60,0	65,0	5
	3	-70,0	-70,5	0,5
	4	-79,0	-74,5	-4,5
CF ₃ CF ₂ OCF ₂ CF ₂ O-		-43,0	-49,5	6,5
$CF_3O(CF_2)_nOCF_2CF_2O-$, where n=	1	-65,0	-74,0	9,0
	2	-74,0	-87,0	13,0
	3	-95,0	-97,0	2,0
	4	-105	-112	7,0
$CF_3(CF_2)_2OCFCF_2O-$ CF_3		-38,0	-40,0	2,0
CF ₃ O(CF ₂) ₃ OCFCF ₂ O- \CF ₃		-54,5	-55,0	0,5
$-(-CF_2CF_2CF_2CF_2-)_n-$				

CF ₃ O-	-18,0	-24,0	6,0
CF ₃ O(CF ₂) ₃ O-	-50,0	-50,0	0
$-(-CFCF_2-)_n-$ R'_F			
CF ₃ O-	-27,0	-38,0	11,0
CF ₃ O(CF ₂) ₃ O-	-56,0	-54,0	-2,0

*for hydrogen-containing polymers the effect of hydrogen bond increment was taken into account in our calculations.

Root mean square deviation of the calculated values from the experimental ones does not exceed +/-6,5°C. In the calculations of T_g for the structure - $[(-Cl_2CF_2)_3CF(R_F)CF_2)_n$ - typical for many fluoroelastomers with ratio 75:25 % mol. vinylidenefluoride and rubber-like co-monomer, such as hexafluoropropylene or various perfluoroalkyl- oxaalkylvinyl ethers, one may apply the empirical formulae based on equations (1, 2):

$$1gT_c = 2.390 - 0.0346 - \sum \frac{1/n_i + 1/m_i}{2}$$

Vitrification temperatures were calculated for the most frostproof heterochain fluoropolymers of various nature, particularly polyfluorosiloxanes and polyfluorophosphazenes.

Table 6. Calculated and experimental vitrification temperatures for some hetero-chain fluoropolymers

CH ₃ -(-SiO-) _n -	T _{g,} ^o C		Δ ±° _C
K F	exper.	calc.	
1	2	3	4
CF ₃ (CH ₂) ₂₋	-73,0	-77,5	4,5
CF ₃ (CF ₂) ₂ O(CF ₂) ₂ (CH ₂) ₂₋	-91,0	-87	-4,0
$CF_3(CF_2)_2OCF(CH_2)_2-$ CF ₃	-89,0	-87,0	-2,0
CF ₃ O(CF ₂) ₃ O(CF ₂) ₂ (CH ₂) ₂₋	-135,0	-136,5	1,5

$\begin{array}{c} R^{*}_{F} \\ -(-P=N-)_{n} - \\ R^{*}_{F} \end{array}$				
CF ₃ CH ₂₋		-65,8	55,8	-10,0
CF ₃ OCF ₂ CF ₂ CH ₂₋		-98,8	-97,3	-1,5
$CF_3O(CF_2O)_nCF_2CH_{2-}$,where n=	2	-140,8	-141,8	1,0
	3	-137,8	-138,8	1,0
	4	-149,8	-150,3	0,5
	5	-153,8	-153,8	0

Root mean square deviation of the calculated value from the experimental one does not exceed +/- 6° C?. To calculate T_g for fluorophosphazene polymers one may apply the empirical equation as follows:

 $\lg T_c = 2,250 - 0,014 - \sum \frac{1/n_i + 1/m_i}{2}$

Thus developed methods for the calculation of T_g values make it possible to forecast the properties of certain fluoropolymers, and forecast T_g values for some novel promising structures. Thus calculated values are shown in Table 7.

Table 7. Calculated T_g values for triazine structures

$\begin{bmatrix} N \\ - R_{F} - C'' & C \\ & & \\ N_{W} & N \\ C & - n \\ & N_{W} & N \\ C & - n \\ CF_{2}O(CF_{2}O)_{7} \end{bmatrix}$	CF3				
-R _F in the main chain	T _{g calc.} ⁰C	-R _F ir chain	the	main	T _{g calc.} ⁰C
1	2	3			4

(- CF ₂ OCF ₂₋) ₂	-133,5	- CF ₂ OCF ₂ OCF ₂ -	-141,0
(- CF ₂ OCF ₂₋) ₃	-129,5	- CF ₂ O(CF ₂ O) ₂ CF ₂₋	-143,5
(- CF ₂ OCF ₂₋) ₄	-126,0	- CF ₂ O(CF ₂ O) ₃ CF ₂₋	-145,5
(- CF ₂ OCF ₂₋) ₅	-123,5	- CF ₂ O(CF ₂ O) ₄ CF ₂₋	-147,0
(- CF ₂ OCF ₂₋) ₆	-121,0	- CF ₂ O(CF ₂ O) ₅ CF ₂₋	-148,5
(- CF ₂ OCF ₂₋) ₇	-119,0	- CF ₂ O(CF ₂ O) ₆ CF ₂₋	-150,0
(- CF ₂ OCF ₂₋) ₈	-117,0	- CF ₂ O(CF ₂ O) ₇ CF ₂₋	-151,0
(- CF ₂ OCF ₂₋) ₉	-115,5		

 $\begin{bmatrix} N \\ - \vdash (CF_2)_{\delta} - C'' \quad C - \vdash \\ | & | & | \\ N_M \quad N & | \\ L & C - R \stackrel{*}{\cdot} J n'$

${\sf R'}_{\sf F}$ in the side chain	T _{g calc.} ^o C	R' _F in the side chain	T _{g calc.} ⁰C
CF ₃ OCF ₂ CF ₂ OCF ₂	-50,5	$CF_3O(CF_2CF_2)_5OCF_2$	-72,5
$CF_3O(CF_2CF_2)_2OCF_2$	-59,0	CF ₃ O(CF ₂ CF ₂) ₆ OCF ₂₋	-75,0
CF ₃ O(CF ₂ CF ₂) ₃ OCF ₂₋	-64,5	CF ₃ O(CF ₂ CF ₂) ₇ OCF ₂₋	-77,5
CF ₃ O(CF ₂ CF ₂) ₄ OCF ₂₋	-65,0	CF ₃ O(CF ₂ CF ₂) ₈ OCF ₂₋	-79,0

「 - → OCFCF ₂) ₃ CF ₂ (CF ₂ OCF) ₃ −C N	$ \begin{array}{c c} \mathbf{N} & \mathbf{V} \\ & \mathbf{V} \\ & \mathbf{V} \\ & \mathbf{V} \\ \mathbf{N} \\ \mathbf{C} - \mathbf{R}_{\mathbf{F}}^{*} \mathbf{J} n^{*} \end{array} $		
CF ₃ OCF ₂ CF ₂ OCF ₂ .	-52,0	CF ₃ O(CF ₂ CF ₂) ₆ OCF ₂₋	-67,0
CF ₃ O(CF ₂ CF ₂) ₃ OCF ₂₋	-59,5	CF ₃ O(CF ₂ CF ₂) ₇ OCF ₂₋	-69,0
CF ₃ O(CF ₂ CF ₂) ₄ OCF ₂₋	-62,5	CF ₃ O(CF ₂ CF ₂) ₈ OCF ₂₋	-71,0
CF ₃ O(CF ₂ CF ₂) ₅ OCF ₂₋	-65,0		
$-(-CH_2CF_2CFCF_2-)_n-*$			
CF ₃ OCF ₂ OOCF ₂ CF ₂ O-	-95,0(-84,5)	CF ₃ O(CF ₂ O) ₅ OCF ₂ CF ₂ O-	-129,0(-123,0)
CF ₃ O(CF ₂ O) ₂ OCF ₂ CF ₂ O-	-108,0(-99,0)	CF ₃ O(CF ₂ O) ₆ OCF ₂ CF ₂ O-	-133,0(-128,5)
CF ₃ O(CF ₂ O) ₃ OCF ₂ CF ₂ O-	-117,(-110,0)	CF ₃ O(CF ₂ O) ₇ OCF ₂ CF ₂ O-	-136,5(-132,5)
CF ₃ O(CF ₂ O) ₄ OCF ₂ CF ₂ O-	-123,5(-118)		
-(-CH ₂ CF ₂) ₃ CFCF ₂ -) _n -* R' _F			
CF ₃ OCF ₂ O-	-67,0	CF ₃ O(CF ₂ O) ₆₋	-118,5
CF ₃ CF ₂ CF ₂ OCF ₂	-27,3	CF ₃ O(CF ₂ O) ₇₋	-123,5
CF ₃ O(CF ₂ O) ₂ O-	-84,0	CF ₃ O(CF ₂ CF ₂ O) ₅₋	-77,5
CF ₃ O(CF ₂ O) ₃ O-	-96,5	CF ₃ O(CF ₂ CF ₂ O) ₆₋	-80,0
CF ₃ O(CF ₂ O) ₄ O-	-105,5	CF ₃ O(CF ₂ O) ₅ OCF ₂ CF ₂ O-	-111,0
CF ₃ O(CF ₂ O) ₅₋	-113,0	CF ₃ O(CF ₂ O) ₆ OCF ₂ CF ₂ O-	-116,5

R_{F}^{\prime}			
CF ₃ O-	-38,0	CF ₃ O(CF ₂ O) ₄ OCF ₂ CF ₂ O-	-126,0
CF ₃ O(CF ₂) ₃ O-	-54,0	CF ₃ O(CF ₂ O) ₅ OCF ₂ CF ₂ O-	-131,5
CF ₃ OCF ₂ OOCF ₂ CF ₂ O-	-92,0	CF ₃ O(CF ₂ O) ₆ OCF ₂ CF ₂ O-	-136,0
CF ₃ O(CF ₂ O) ₂ OCF ₂ CF ₂ O-	-107,5	CF ₃ O(CF ₂ O) ₇ OCF ₂ CF ₂ O-	-139,5
CF ₃ O(CF ₂ O) ₃ OCF ₂ CF ₂ O-	-118,0		

(-CECE--)

R'F			
CF ₃ O-	-24,0	CF ₃ O(CF ₂ O) ₄ OCF ₂ CF ₂ O-	-111,0
CF ₃ O(CF ₂) ₃ O-	-42,0	CF ₃ O(CF ₂ O) ₅ OCF ₂ CF ₂ O-	-118,0
CF ₃ OCF ₂ OOCF ₂ CF ₂ O-	-74,5	CF ₃ O(CF ₂ O) ₆ OCF ₂ CF ₂ O-	-123,0
CF ₃ O(CF ₂ O) ₂ OCF ₂ CF ₂ O-	-90,5	CF ₃ O(CF ₂ O) ₇ OCF ₂ CF ₂ O-	-127,50
CF ₃ O(CF ₂ O) ₃ OCF ₂ CF ₂ O-	-102,5		

-(-CF2CF2CF2-)n-

K F			
CF ₃ O-	-62,0	CF ₃ O(CF ₂ O) ₄₋	-129,0
CF ₃ O(CF ₂) ₃ O-	-64,5	CF ₃ O(CF ₂ O) ₅₋	-135,0
CF ₃ OCF ₂ O-	-93,0	CF ₃ O(CF ₂ O) ₇₋	-139,0
CF ₃ O(CF ₂ O) ₂₋	-110,0	CF ₃ O(CF ₂ O) ₈₋	-142,5
CF ₃ O(CF ₂ O) ₃₋	-121,5		

-(-SiO-)n- R'F			
CF ₃ OCF ₂ O(CF ₂) ₂ (CH ₂) ₂₋	-123,0	CF ₃ O(CF ₂) ₅ O(CF ₂) ₂ (CH ₂) ₂ .	-144,0
CF ₃ O(CF ₂) ₂ O(CF ₂) ₂ (CH ₂) ₂ .	-131,0	CF ₃ O(CF ₂) ₆ O(CF ₂) ₂ (CH ₂) ₂₋	-147,0
CF ₃ O(CF ₂) ₄ O(CF ₂) ₂ (CH ₂) ₂ .	-141,0	CF ₃ O(CF ₂) ₇ O(CF ₂) ₂ (CH ₂) ₂₋	-149,0
/R' _F -(-SiO-) _n - R' _F			
CF ₃ OCF ₂ O(CF ₂) ₂ (CH ₂) ₂₋	-114,0	CF ₃ O(CF ₂) ₅ O(CF ₂) ₂ (CH ₂) ₂ .	-142,0
CF ₃ O(CF ₂) ₂ O(CF ₂) ₂ (CH ₂) ₂ .	-125,5	CF ₃ O(CF ₂) ₆ O(CF ₂) ₂ (CH ₂) ₂₋	-145,5
CF ₃ O(CF ₂) ₃ O(CF ₂) ₂ (CH ₂) ₂ .	-132,5	CF ₃ O(CF ₂) ₇ O(CF ₂) ₂ (CH ₂) ₂ .	-148,0
CF ₃ O(CF ₂) ₄ O(CF ₂) ₂ (CH ₂) ₂ .	-138,0		
/R [°] _F -(-P=N-) _n - 'R [°] _F			
CF ₃ O(CF ₂) ₆ O(CF ₂) ₂ (CH ₂) ₂ .	-160,3	CF ₃ O(CF ₂) ₈ O(CF ₂) ₂ (CH ₂) ₂₋	-161,8
CF ₃ O(CF ₂) ₇ O(CF ₂) ₂ (CH ₂) ₂ .	-156,8	CF ₃ O(CF ₂) ₉ O(CF ₂) ₂ (CH ₂) ₂ .	-160,8

Therefore, reasoning from the mentioned experimental and calculated results one may conclude that the pattern of calculation proposed for vitrification temperature makes it possible to forecast rather satisfactorily freezeproof characteristics of various carbon-chain and heterochain fluoropolymers in order to select the promising materials with the optimal properties. Apparently, among the investigated materials polyfluorooxamethylenephosphazene polymers have the best freeze resistance, as for them T_gvalues are close to its limit value - 170° C.

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CH₃

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